A Single-Crystal Diffractometry Investigation of Iron in β-Rhombohedral Boron

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The structure of a crystal of composition FeB_{~49} has been investigated using single-crystal diffractometry. The space group is $R^{3}m$ and the hexagonal cell dimensions are a = 10.951 Å and c = 23.861 Å. The three-dimensional boron network is essentially equivalent to that found in β -rhombohedral boron. Two sets of holes in this network are partially occupied by iron atoms.

Introduction

Determination of the structure of the solid solution of chromium in β -rhombohedral boron (1) has revealed that the chromium atoms partially occupy two sets of holes in the β -rhombohedral boron network and leave at least one set empty, which is large enough to accommodate chromium atoms. Similar solid solutions of the remaining 3d elements have been observed using X-ray powder data, and the cell expansion was measured and correlated with the microhardness (2, 3). Recently, an investigation of the solubility of copper and manganese in β -rhombohedral boron (4) showed that atoms of these metals are distributed among three different sets of interstitial positions in the β -boron structure. It is evident that the different behavior of chromium, copper, and manganese in β -boron cannot be explained solely in terms of size factor restrictions, since copper and chromium have very closely equal atomic radii, so it seemed of interest to study how other transition metals atoms are distributed when dissolved in β boron. Two Mössbauer studies of iron in β boron have been reported, but even allowing for the difference in the sample preparation techniques used in the two investigations, the results obtained seem contradictory. In one of the investigations (5) it was concluded that iron, when dissolved in β -boron, occupies

three different crystallographic positions. As a result of the other investigation (6) it was proposed that iron is substitutionally dissolved in β -boron. It was felt desirable to relate these results to a fundamental crystallographic study

Experiments and Crystal Data

Crystalline boron (Borax Consolidated, claimed purity 99.8 %) and iron rods (Johnson Matthey Chemicals Ltd., specpure) were arcmelted under an atmosphere of purified argon to a nominal composition corresponding to $FeB_{10,3}$. The crystal investigated was taken from this sample. The cell dimensions were measured for the single crystal as well as the solidified melt, using previously described techniques (1, 4). From powder diffraction patterns it was established that the sample of nominal composition FeB_{10.3} contained two phases, namely, FeB and a phase similar to the CuB~28 and MnB~23 phases previously reported (4). From the single-crystal refinement, the latter phase was found to have the composition $FeB_{\sim 49}$. The hexagonal cell dimensions of this phase were: from powder, a =10.9632(15) Å; c = 23.867(4) Å; from single crystal, a = 10.9514(8) Å; c = 23.861(2) Å. The agreement between the two values for the caxis is acceptable while the *a*-axis of the single crystal is slightly shorter than that of the

Copyright © 1976 by Academic Press, Inc. All rights of reproduction in any form reserved. Printed in Great Britain powder, and the single crystal may thus contain less iron than the main part of the sample. However, the two cell dimensions measured are substantially larger than those of pure β -boron, which are a = 10.9253(5) Å and c = 23.8103(16) Å (1).

Intensities were recorded with an automatic STOE-Philips four-circle diffractometer using graphite monochromatized MoK α radiation and the $\omega/2\theta$ scanning technique. The background was measured on each side of the scanning interval, which was 1.60°. All reflections up to $2\theta = 60^\circ$ with $h \le 0$, $k \ge 0$, and $l \ge 0$ were measured. The crystal had a maximum cross section of 0.12 mm and the calculated linear absorption coefficient for MoK α radiation was 10 cm⁻¹. Weissenberg films were recorded to check the crystal quality and the symmetry.

The crystallographic calculations were performed using IBM 370/155 and IBM 1800 computers. A list of the programs used is given in (4). The atomic scattering factors, taken taken from (7), were corrected for anomalous dispersion according to Cromer (8).

Refinement of the Structure

The original set of data consisted of 1870 intensities, which were corrected for Lorentz and polarization effects. The F_0^2 values for equivalent reflections were averaged, after which the data were reduced to 919 independent structure factors, which included all observed reflections with $2\theta \leq 60^\circ$ (Table I).

Using the coordinates and the temperature factors of the boron atoms in $\operatorname{CrB}_{-41}(1)$, threedimensional F_0 syntheses were calculated. In these syntheses peaks were found at the positions A(1) and D(1) (notation according to (1), the corresponding atoms are denoted Fe(1) and Fe(2) below). Approximate atomic coordinates and occupational parameters for the iron atoms were estimated from the F_0 and F_c syntheses. In the subsequent refinements the

นองหารทระเอราใช้ระวิธาณีขึ้นที่มีสามารถและเกิดรายตรงรารทระเอรายังกับราที่สองและรางได้ 1. 1. 1. 1. 1. 1. 1. 1. 1 เป็นประวัติเป็นที่มีสามารถใช้ระเอราเรียงเป็นจากกับรายตรงรารทระเอราย์ที่สองและรางการสามารถตรงรายสามารถหลาย ราย
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TABLE I

Observed and Calculated Structure Factors (\times 10) for FeB_{~49}

least-squares program XTLALS (9) was used to refine one scale factor, the atomic coordinates, anisotropic, or isotropic temperature factors, and occupational parameters for the iron atoms and one boron atom. The function minimized was $\sum w(|F_0|^2 - |F_c|^2)^2$. Weights were used according to $w = 1/{\{\sigma^2(F_0)^2\}} + (C \cdot F_0^2)^2\}$ with $\sigma(F_0^2)$ obtained from counting statistics. The best weighting scheme was obtained with C = 0.045. The agreement index given below was defined as $R = \sum (|F_0| - |F_c|)/\sum |F_0|$.

Three different refinements were performed, namely, an anisotropic refinement using all 919 observed reflections, and an isotropic as well as an anisotropic refinement using a data set of 765 reflections (reflections with $F_0{}^2 < 2\sigma(F_0{}^2)$ were deleted and were marked with a dot in Table I). The type of refinement, the number of reflections used, and the *R*-values obtained were as follows: anisotropic, 765 reflections, 4.3%; isotropic, 765 reflections, 4.4%; anisotropic, 919 reflections, 5.4%.

The *R*-values obtained were lower for the smaller data set and the standard deviations of the atomic positions were somewhat larger, the increase being 6-9% for the anisotropic and 4-7% for the isotropic refinement as compared with the anisotropic refinement using 919 reflections. The atomic coordinates of the isotropic refinement using 765 reflections differed very little from those obtained in the anisotropic refinement of 919 reflections. The maximum difference was that found for the y coordinate of the partially occupied Fe(2)position. It was two-thirds of the standard deviation (0.0009 Å). The differences in the occupational parameters were also very small. The composition as calculated from the isotropic refinement (765 reflections) was FeB48.81 and FeB_{48.71} from the anisotropic one (919 reflections).

Atom	Position	x	У	Z
B(1)	36(<i>i</i>)	0.17503(19)*	0.17544(19)	0.17689(7)
B(2)	36(<i>i</i>)	0.31875(19)	0.29492(19)	0.12892(7)
B(3)	36(<i>i</i>)	0.26155(19)	0.21724(19)	0.41965(7)
B(4)	36(<i>i</i>)	0.23580(18)	0.25150(19)	0.34683(7)
B(5)	18(<i>h</i>)	0.05465(13)	0.10930(13)	0.94412(10)
B(6)	18(<i>h</i>)	0.08596(13)	0.17192(13)	0.01329(10)
B(7)	18(<i>h</i>)	0.11048(14)	0.22095(14)	0.88704(10)
B(8)	18(<i>h</i>)	0.16968(14)	0.33936(14)	0.02790(10)
B(9)	18(<i>h</i>)	0.12900(14)	0.25801(14)	0.76628(10)
B(10)	18(<i>h</i>)	0.10226(14)	0.20451(14)	0.69834(10)
B(11)	18(<i>h</i>)	0.05651(13)	0.11301(13)	0.32662(10)
B(12)	18(<i>h</i>)	0.08991(14)	0.17983(14)	0.39848(10)
B(13) ^c	18(<i>h</i>)	0.05641(21)	0.11282(21)	0.55487(16)
B(14)	6(c)	0	0	0.38550(19)
B(15)	3(b)	0	0	1/2
$Fe(1)^d$	6(<i>c</i>)	0	0	0.13528(4)
Fe(2) ^e	18(<i>h</i>)	0.20235(11)	0.40470(11)	0.17302(8)

TABLE II

STRUCTURE	DATA	FOR	FeB~4	9 ^a
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^{*a*} Space group $R^{3}m$ (no. 166). Hexagonal unit-cell dimensions: a = 10.9514(8) Å, c = 23.8609(16) Å. Cell volume: 2478.3 Å³. (Rhombohedral unit cell: a = 10.161 Å, $\alpha = 65.22^{\circ}$, V = 826.1 Å³.)

^b Standard deviations are given in parentheses and refer to the last decimal place of the respective values.

^c Occupancy 72.6(11)%.

⁴ Occupancy 50.7(3)%.

^e Occupancy 18.5(2)%.

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THERMAL PARAMETERS FOR $FeB_{\sim 49}$ (in Å²)

Atom	\mathbf{B}_{11}	B ₂₂	B ₃₃	B ₁₂	B ₁₃	B ₂₃	В
B(1)	0.517(63)ª	0.538(65)	0.492(59)	0.300(52)	0.032(48)	0.033(48)	0.499
B(2)	0.361(60)	0.444(62)	0.383(57)	0.174(50)	-0.055(44)	-0.035(46)	0.408
B(3)	0.399(62)	0.510(64)	0.517(64)	0.266(52)	-0.030(46)	-0.042(45)	0.458
B (4)	0.364(61)	0.472(62)	0.403(59)	0.188(51)	-0.031(45)	-0.036(47)	0.423
B(5)	0.300(62)	0.300(62)	0.519(84)	0.143(71)	-0.002(33)	0.002(33)	0.376
B (6)	0.289(59)	0.289(59)	0.385(84)	0.101(69)	-0.001(33)	0.001(33)	0.340
B (7)	0.318(63)	0.318(63)	0.522(82)	0.101(71)	-0.062(33)	0.062(33)	0.412
B(8)	0.426(63)	0.426(63)	0.405(89)	0.193(71)	0.042(33)	0.042(33)	0.428
B (9)	0.398(64)	0.398(64)	0.487(84)	0.195(73)	0.018(33)	-0.018(33)	0.430
B (10)	0.402(65)	0.402(65)	0.301(80)	0.212(72)	-0.024(32)	0.024(32)	0.363
B (11)	0.443(65)	0.443(65)	0.369(82)	0.277(74)	0.014(33)	-0.014(33)	0.394
B(12)	0.458(63)	0.458(63)	0.241(80)	0.228(72)	0.009(32)	-0.009(32)	0.386
B (13)	0.97 (12)	0.97 (12)	0.88 (15)	0.55 (12)	-0.046(53)	0.046(53)	0.912
B(14)	0.435(93)	0.435(93)	0.35 (14)	0.22 (5)	0	0	0.408
B(15)	0.79 (14)	0.79 (14)	0.09 (20)	0.392(72)	0	0	0.554
Fe(1)	0.257(31)	0.257(31)	0.210(41)	0.128(15)	0	0	0.241
Fe(2)	0.899(64)	0.899(64)	0.979(84)	0.397(63)	-0.031(28)	0.031(28)	0.950

^a Standard deviations are given in parentheses and refer to the last decimal place of the respective values. B_{ij} is defined as $8\pi^2 U_{ij}$ where U_{ij} is the mean-square displacement tensor.

Evidently, the results of the three refinements differ very little. The decrease of the Rvalue from 0.044 to 0.043, when anisotropic thermal parameters were introduced, is insignificant. Although the reduction of the data set from 919 to 765 reflections involves a decrease in the R value from 0.054 to 0.043, the standard deviations increase. We conclude that all 919 structure factors should be used in the refinement, since this gives the lowest standard deviations. The structure data, presented in Tables II and III, are based on 919 structure factors and anisotropic thermal parameters.

The two iron atoms, Fe(1) and Fe(2), were included in a final three-dimensional difference synthesis over the whole asymmetric unit. The largest deviation from zero for the electron density was 5% of a boron maximum in the F_0 synthesis. Consequently, it was concluded that no further iron atoms are found in the structure.

Discussion

The crystal structure of β -rhombohedral boron was described and discussed in detail

by Hoard et al. (10, 11). Brief descriptions of the structure as well as the system of holes occurring in the structure were given later (1, 4). Therefore, only a few remarks regarding the specific situation in $\text{FeB}_{\sim 49}$ will be given below.

A comparison of the atomic coordinates of the boron atoms in pure β -boron with those in FeB_{~49} reveals no great differences between the two structures. The three-dimensional boron network is essentially the same, and the largest deviation in boron-boron distance between the two structures occurs for the B(13)-B(14) distance, which is 0.032 Å shorter in FeB_{~49}.

The Fe(1) atoms are symmetrically surrounded by 12 boron nearest neighbors at a distance of 2.16 Å (Table IV), which is very close to the sum of the atomic radii of boron (0.88 Å) and of iron (1.27 Å). The Fe(1) atoms have no close metal neighbors. The Fe(2) atom have 14 boron neighbors at distances between 2.28 and 2.49 Å. The very short Fe(2)–B(13) distance (Table IV) is of rare occurrence, since occupancy of the two positions is low, namely, 19% for Fe(2) and 73%

TABLE IV

B(1)-B(9)	1.840(3) ^a	B(6)-B(8)	1.626(3)	B(13)-B(15)	1.691(4)
-B(2)	1.842(3)	-2 B (6)	1.750(3)	-B(14)	1.780(5)
-B(2)	1.855(2)	-2 B (5)	1.754(2)	-2B(3)	1.804(4)
-B(7)	1.856(3)	-B(5)	1.754(3)	-2B(13)	1.853(6)
-B(1)	1.912(4)			-2B(12)	1.862(3)
-B(1)	1.926(4)			-Fe(2)	2.033(4)
-Fe(1)	2.161(2)	B(7) - B(5)	1.725(3)	-2Fe(2)	2.449(3)
-Fe(2)	2.377(2)	-2B(2)	1.791(2)		
		-B(9)	1.809(3)		
		-2B(1)	1.856(3)	B(14) - 3B(12)	1.733(2)
B(2) - B(3)	1.729(2)	-Fe(1)	2.162(2)	- 3B (11)	1.767(4)
-B(7)	1.791(2)			-3B(13)	1.780(5)
-B(2)	1.822(3)				
- B (1)	1.842(3)	B(8)-B(6)	1.626(3)		
-B(9)	1.849(2)	-2B(4)	1.736(2)	B(15)-6B(13)	1.691(4)
-B(1)	1.855(2)	-2B(3)	1.797(3)	-6Fe(2)	2.484(2)
-Fe(2)	2.390(2)	-B (10)	1.838(3)		
-Fe(2)	2.419(2)				
		B(9)–B(10)	1.699(3)	Fe(1) - 3B(5)	2.160(3)
B(3) - B(2)	1.729(2)	-B(7)	1.809(3)	-6B(1)	2.161(2)
-B(12)	1.785(2)	-2B(1)	1.840(3)	-3B(7)	2.162(2)
- B (8)	1.797(3)	-2B(2)	1.849(2)	-3Fe(2)	3.943(2)
-B(13)	1.804(4)				
-B(4)	1.829(2)	B(10)-B(9)	1.699(3)	Fc(2)-B(13)	2.033(4)
-B(3)	1.894(4)	-2B(11)	1.785(2)	-2B(3)	2.275(2)
-Fe(2)	2.275(2)	- B (8)	1.838(3)	-2B(1)	2.377(2)
-Fe(2)	2.470(2)	-2B(4)	1.841(2)	-2B(2)	2.390(2)
				-B(12)	2.401(3)
B(4)-B(4)	1.681(3)			-2B(2)	2.419(2)
-B(8)	1.736(2)	B(11)–B(14)	1.767(4)	-2B(13)	2.449(3)
-B(3)	1.829(2)	-2B(10)	1.785(2)	-2B(3)	2.470(2)
-B(10)	1.841(2)	-B(12)	1.828(3)	-B(15)	2.489(2)
- B (11)	1.847(2)	-2B(4)	1.847(2)	-2Fe(2)	2.503(2)
-B(12)	1.853(2)	-2B(11)	1.856(4)	-Fe(1)	3.443(2)
B(5)-B(7)	1.725(3)	B(12)-B(14)	1.733(2)		
-2B(6)	1.754(2)	-2B(3)	1.785(2)		
-B(6)	1.754(3)	-B(11)	1.828(3)		
-2B(5)	1.795(4)	-2B(4)	1.853(2)		
-Fe(1)	2.160(3)	-2B(13)	1.862(3)		
		-Fe(2)	2.401(3)		

INTERATOMIC DISTANCES IN FEB~49 (IN Å)

" Standard deviations in parentheses. Distances listed at B-B <2.70 Å, Fe-B <3.0 Å, and Fe-Fe <4.0 Å.

for B(13). Full occupation of the Fe(2) positions would involve the formation of iron chains in the structure with iron-iron atoms distances of 2.50 Å. The structure determination of pure β -boron (10) revealed a partial occupation of the B(13) and B(16) positions. The occupancy of these positions as well as that of the metal positions have been collected in Table V for the solid solutions of chromium, manganese, copper, and iron in β boron. It is evident from Table V that iron closely resembles chromium in its distribution

		Occupation (%)					
Atom	$\operatorname{Cr} B_{\sim 41}(l)$	$MnB_{\sim 23}(4)$	$\operatorname{CuB}_{\sim 28}(4)$	FeB~49	β-boron (10)		
Me(1)	71.9(6)	25.6(7)	6.1(5)	50.7(3)	······		
Me(2)	18.0(3)	43.1(5)	43.1(6) ^a	18.5(2)			
Me(3)	0	66.2(8)	50.5(6)	0			
B(13)	71.7(18)	65.0(23)	61.1(19)	72.6(11)	~67		
B (16)	0	0	20.9(22)	0	~33		

TABLE	V
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" The value refers to the sum of Cu(2') in 18(h) and Cu(2'') in 36(i).

among the various sets of holes in the β -boron structure. The occupancy at the Me(1) positions is, however, lower for the iron atoms. Table V also shows that the occupation of the B(13) position is fairly independent of the introduced metal atom, while the occurrence of B(16) atoms seems to be coupled to a low occupation of the Me(1) position.

At the end of this investigation we became aware of a structure refinement of a crystal with the composition $FeB_{29,5}$ undertaken by Hughes and Tai at Cornell University (12). They found iron at the same atomic positions as we report here, although occupancy was higher at both positions due to the higher iron content of their crystal. Thus, it has been established in two independent investigations that iron dissolves interstitially in β -boron and that it is distributed between two different crystallographic positions. In view of this evidence it is impossible to accept the proposal by Stanke and Parak (6), based on a Mössbauer study, that iron is substitutionally accommodated in the β -boron structure.

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